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Structural calculations and phase stability

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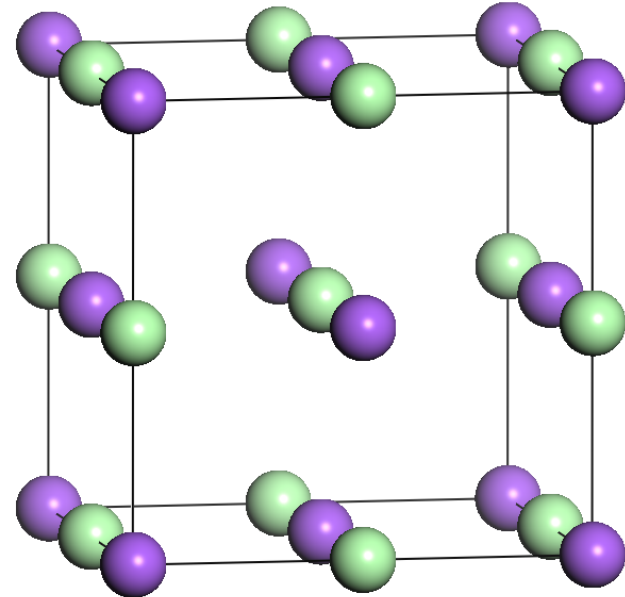
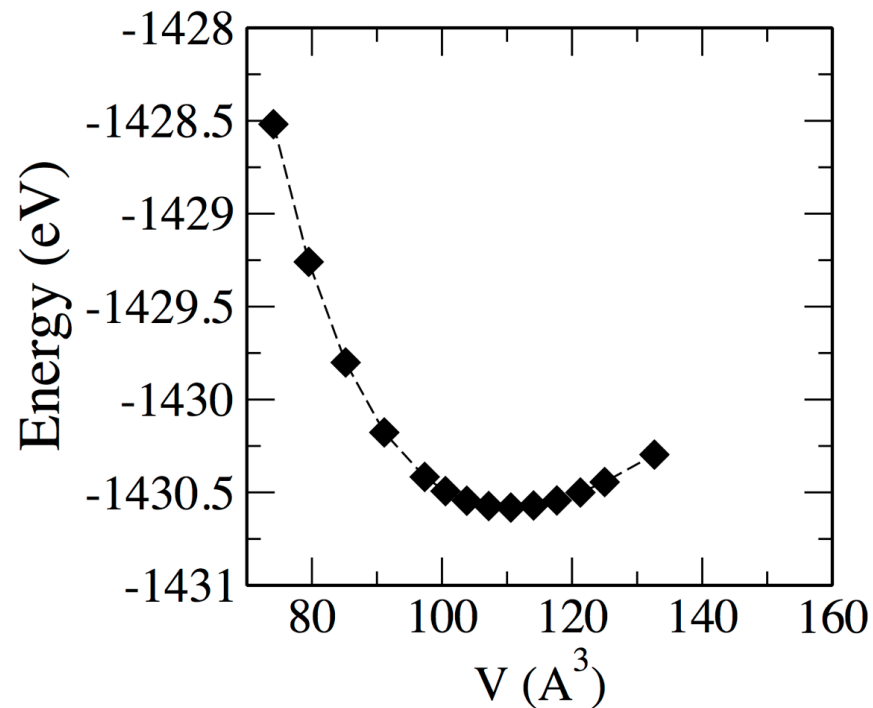
- Total Energy
 - Lattice constants
 - Pressure driven phase transitions
 - Bond energy, formation energy, surface ...
- Forces
 - Hellman-Feynman theorem
- Stresses
- Elastic constants

Total Energy

- Quantum Mechanics gives a complete description of materials chemistry & physics
 - QM of electrons and nuclei.
- Covalent, ionic, metallic and van der Waals bonding are all emergent properties of the QM behaviour of electrons.
- Born-Oppenheimer Approximation
 - Treat nuclei classically and assume electrons move adiabatically in a field of fixed ions.
 - QM electrons give interactions between atoms

-
- Primary computed property in DFT is the ground state energy.
 - All observables related to the *change* in energy when the system is perturbed.
 - Crystal structure
 - Ion positions and lattice parameters that minimize the total energy.
 - Phase stability
 - Phase with lowest E is most stable at $T=0$
 - Elastic constants

- Consider a simple structure:
- The equilibrium CaO lattice parameter is given by the minimum energy:



- A popular EoS is the Birch-Murnaghan:

$$E(V) = E_0 + \frac{9V_0 B_0}{16} \left\{ \left[\left(\frac{V_0}{V} \right)^{\frac{2}{3}} - 1 \right]^3 B'_0 + \left[\left(\frac{V_0}{V} \right)^{\frac{2}{3}} - 1 \right]^2 \left[6 - 4 \left(\frac{V_0}{V} \right)^{\frac{2}{3}} \right] \right\}$$

- Where the equilibrium volume is V_0 and the bulk modulus is given by

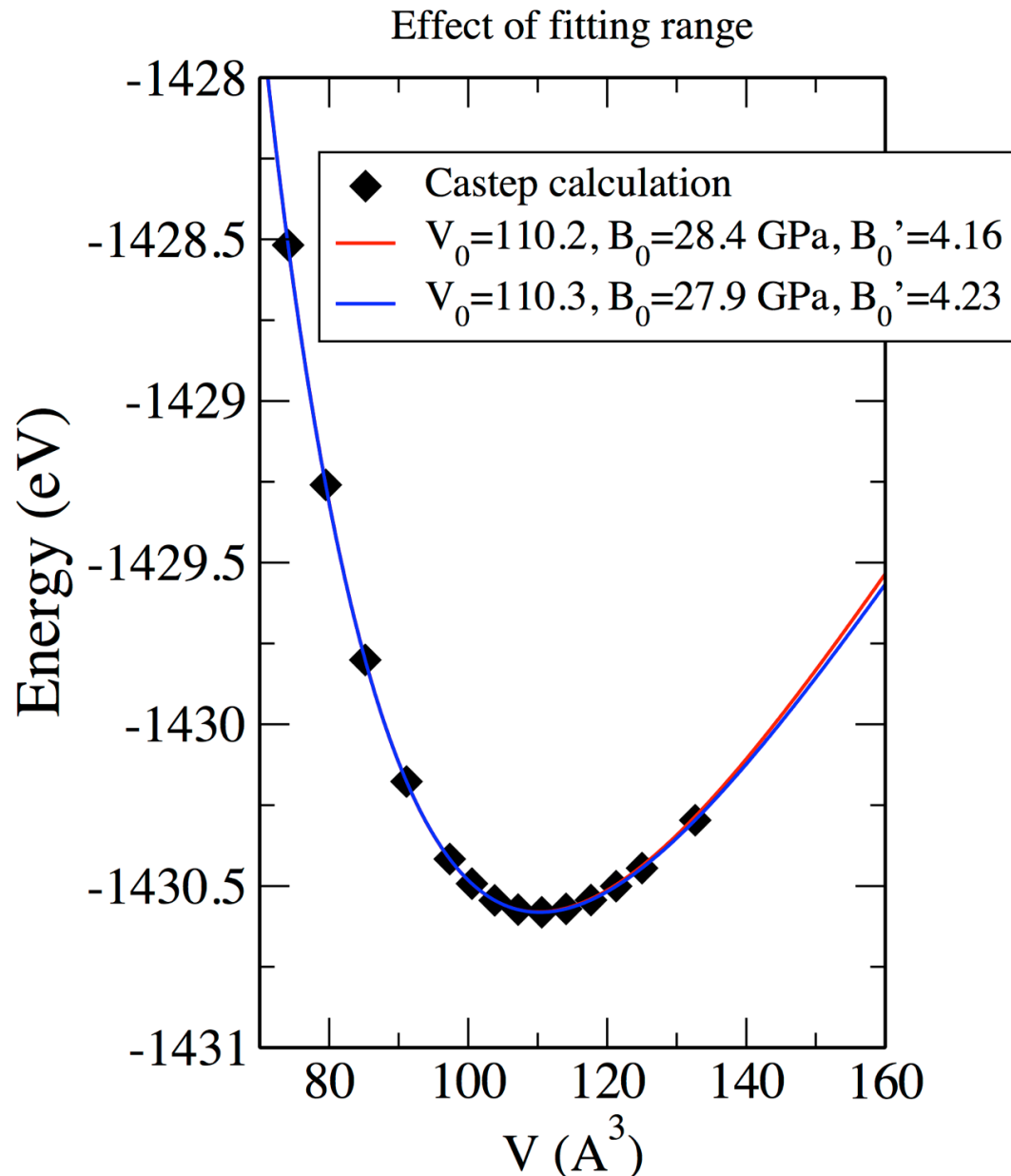
$$B_0 = -V \left. \frac{\partial P}{\partial V} \right|_{P=0}$$

- And the pressure derivative is $B'_0 = \left. \frac{\partial B}{\partial P} \right|_{P=0}$

- The bulk modulus is a physical observable
- But two definitions
 - Can extract from fitting B-M EoS to compression data
 - Or from 2nd derivative as in ultrasound experiments

$$B_0 = -V_0 \frac{d^2 P}{dV^2}$$

- And the results are not the same! Why?

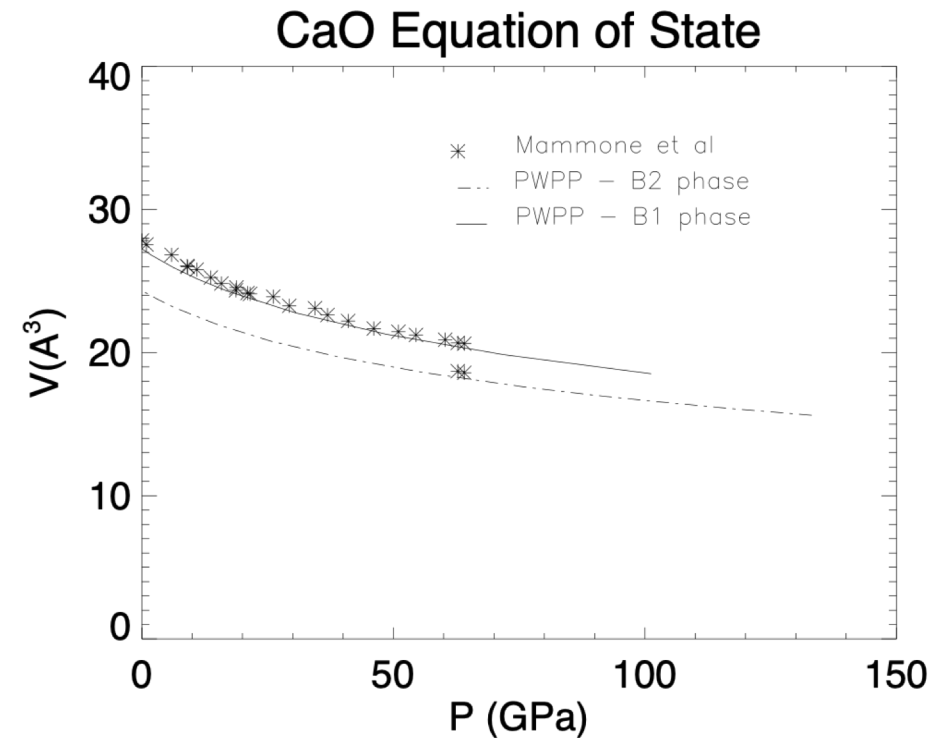
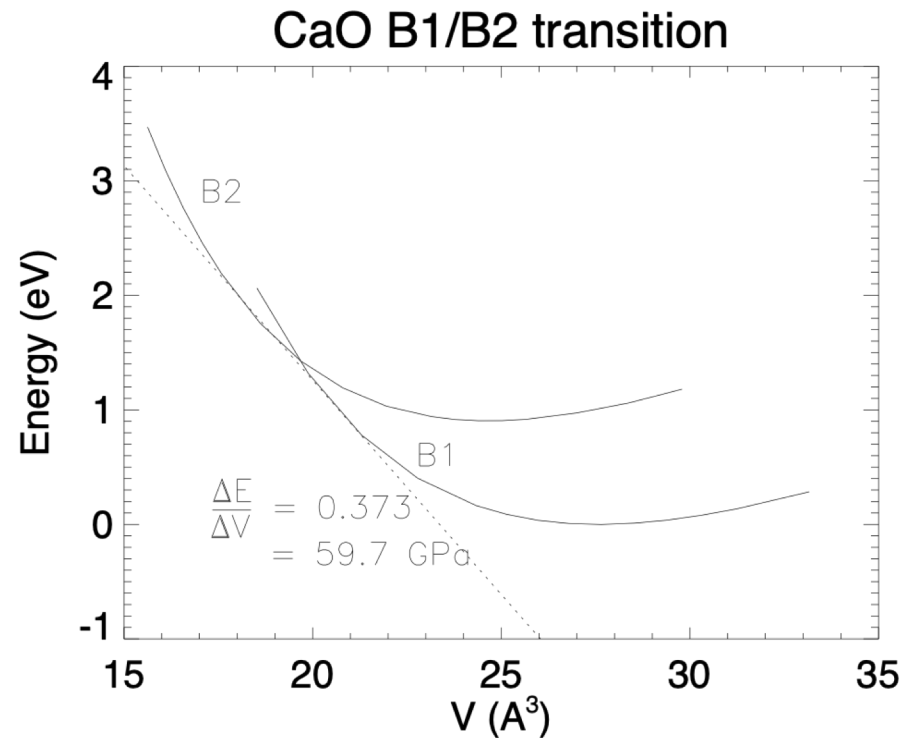


- Range of data changes fit!
- Experiment can only do $V \leq V_0$ but calculation can do both
- Must use same (or similar) range of V as in experiment to get comparable B_0

- A structure will change phase to the one of lowest free energy
- Phase equilibrium happens if:
 - Pressure of both phases are equal
 - Enthalpy of both phases are equal:

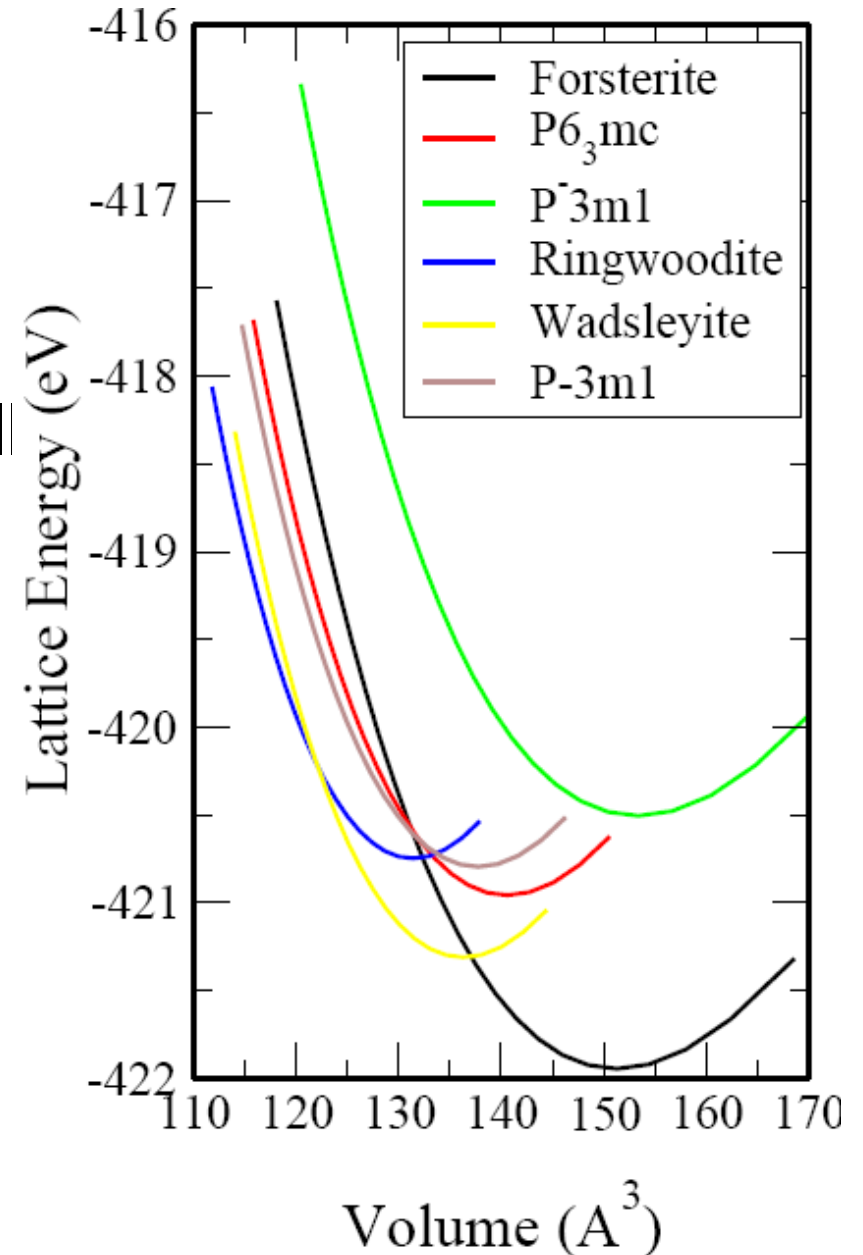
$$E_2 - E_1 = P(V_2 - V_1)$$

- Can use gradient of common tangent to get transition pressure ...

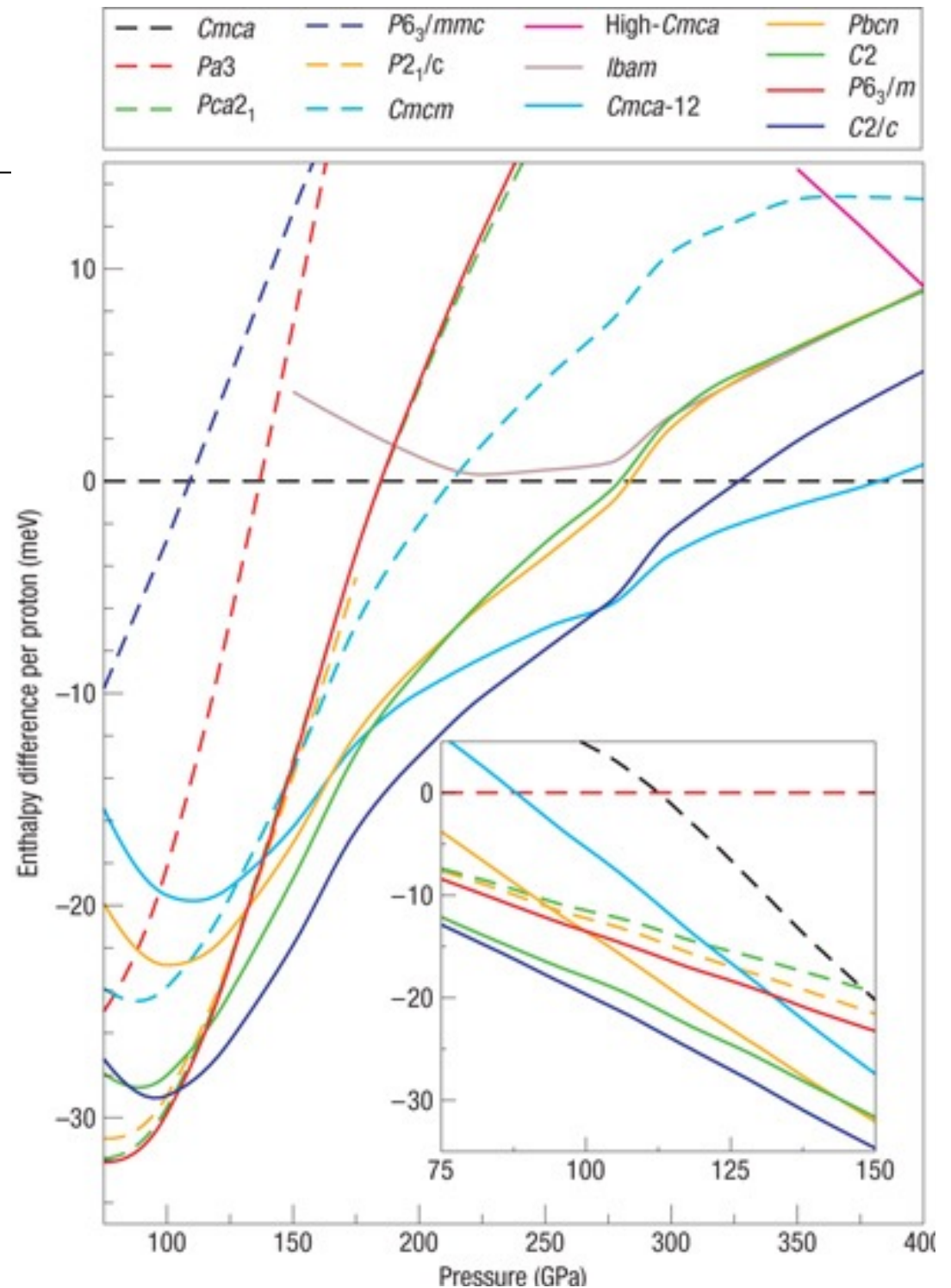


- Or can plot enthalpy H vs pressure P and see which phase has lowest H at given P
- Requires accurate P calculation ...

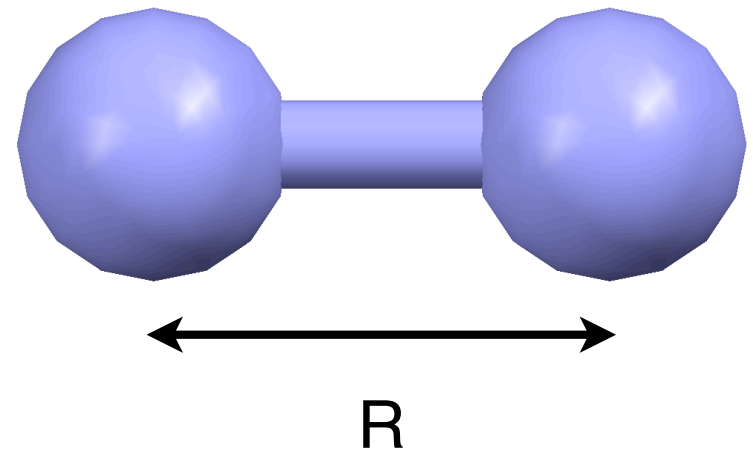
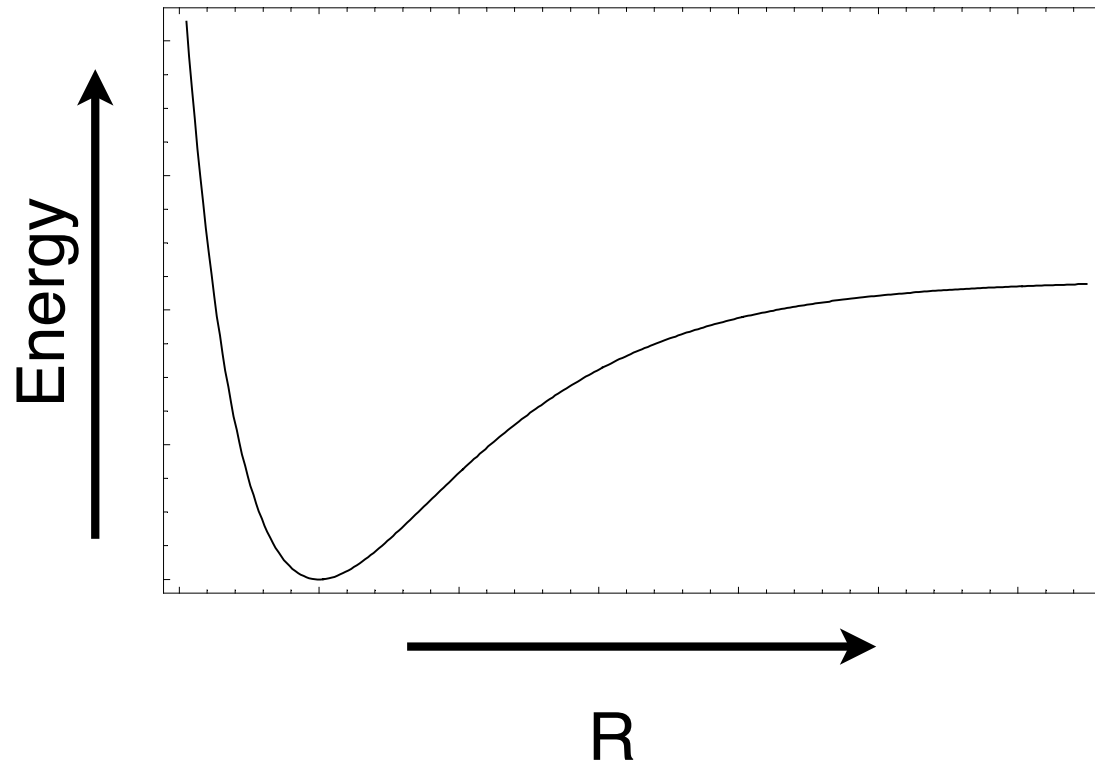
- Which phase is most stable?
 - calculate $E(V)$ curve
 - Forsterite is most stable of all the phases considered
 - then Wadsleyite and then Ringwoodite as increase pressure
- Limited to set of considered structures
 - What if miss something?
 - Active research topic ...



- High pressure Hydrogen
- Plot ΔH wrt *Cmca* phase
- Solid lines are new structures, dashed had been already considered by others



- Atoms form bonds in solids in molecules
- Energy varies with length of bond:



- Hence min energy = equilibrium bond length

- If we stretch a bond too much it will break
 - Energy required is the *dissociation energy*, i.e. $\Delta H = E(\text{molecule}) - \text{sum } E(\text{fragments})$
 - Example: methane CH_4
 - dissociation energy = $E(\text{CH}_4) - E(\text{CH}_3) - E(\text{H})$
 - Covalent bonds are strong – typically 1-5 eV
 - Environment dependent
- Always requires energy => endothermic
 - $\text{H}_3\text{C-H}$ bond is ~4.5 eV as bond is short
 - N_2 triple bond is VERY strong = 9.8 eV

- Average dissociation energy over all bonds in a molecule is the *bond energy*
 - More transferable
 - Example: methane CH₄
 - bond energy = $\frac{1}{4} * (E(\text{CH}_4) - E(\text{C}) - 4E(\text{H}))$
- NB long bonds are weak
 - BE of H-H = 4.5 eV, length = 0.74 Å
 - BE of H-C = 4.3 eV, length = 1.09 Å
 - BE of H-Si = 3.3 eV, length = 1.48 Å
 - BE of Si-Si = 2.3 eV, length = 2.33 Å

- For solids or liquids
 - The energy required to break all the bonds between the basic units (atoms or molecules) of the material
 - i.e. the intra-molecular bonds
- Example: silicon
 - Cohesive energy = $1/n * (E(\text{Si}_n) - n * E(\text{Si}))$
- Example: water ~ 0.4 eV/molecule
 - Relates to boiling point of water – not dissociating water into H_2 and $1/2 \text{O}_2$!

- What about defects? Simplest is the point vacancy – a single missing atom ...
- Neutral vacancy in an elemental solid:

$$E_V = E_{N-1} - \left(\frac{N-1}{N} \right) E_N$$

- Need to consider effect of relaxation around the vacancy – inward or outward?
- Might also be symmetry breaking
 - e.g. Jahn-Teller effect

- More general, for charged defects in non-elemental system:

$$E_{\text{form}} = G_{\text{defect}} - G_{\text{ideal}} = H_{\text{defect}} - H_{\text{ideal}} + \sum_j n_j \mu_j^{\text{removed}} - \sum_j n_j \mu_j^{\text{added}} - n_e \mu_e$$

- Includes change in free energy and change in chemical potential and charge transfer
- Depends on reservoir
 - Where do the defect atoms go/come from?
- Neglects change in entropy

- For an elemental solid, chemical potential μ is the energy of gas phase
- For an compound solid, depends on chemical bonds present
 - E.g. quartz (SiO_2):

$$\mu_{\text{SiO}_2} = \mu_{\text{Si}} + \mu_{\text{O}_2} = \mu_{\text{Si}}^* + \mu_{\text{O}_2}^* + \Delta H_{\text{SiO}_2}$$

$$\mu_{\text{Si}}^* + \Delta H_{\text{SiO}_2} \leq \mu_{\text{Si}} \leq \mu_{\text{Si}}^*$$

$$\frac{1}{2}\mu_{\text{O}_2}^* + \frac{1}{2}\Delta H_{\text{SiO}_2} \leq \mu_{\text{O}} \leq \frac{1}{2}\mu_{\text{O}_2}^*$$

where μ_{Si}^* is chem pot of gas-phase Si, and $\mu_{\text{O}_2}^*$ is the chem pot of diatomic gas-phase O_2

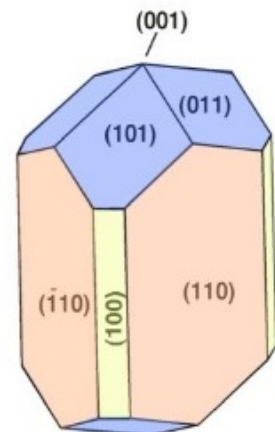
- As well as studying bulk materials and defects, can also study surfaces
- Surface energy γ is the energy required to form a surface:

$$\gamma = \frac{1}{2A} \left(E_{surface} - k * E_{bulk} \right)$$

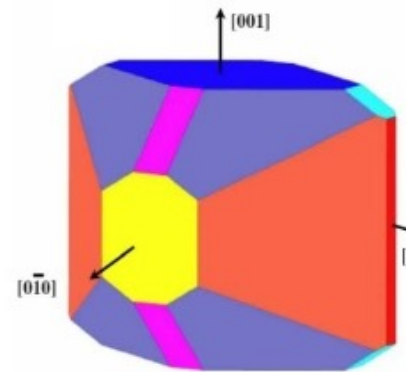
- Where the surface calculation contains 2 surfaces separated by k planes of atoms
- And E_{bulk} is the energy in bulk of 1 plane
- And A is the area of the surface created
- Can also be measured directly in expt

- The equilibrium shape of a crystal or nanoparticle is given by min Gibbs free energy
- Surface Gibbs free energy is $\Delta G = \sum_j \gamma_j A_j$
 - Variational problem – minimize ΔG subject to fixed volume constraint
 - Hence surface with lowest γ will dominate

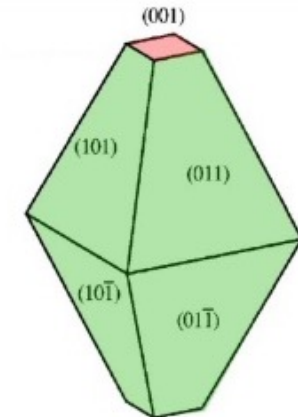
- e.g. TiO_2



rutile
 $\langle \gamma \rangle = 1.1 \text{ J/m}^2$



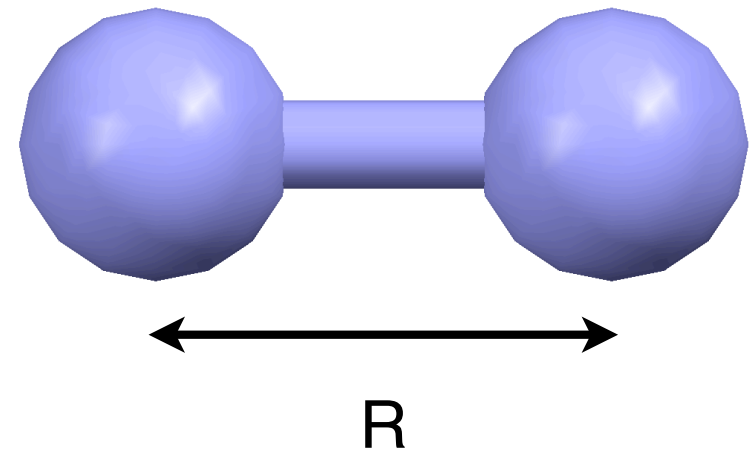
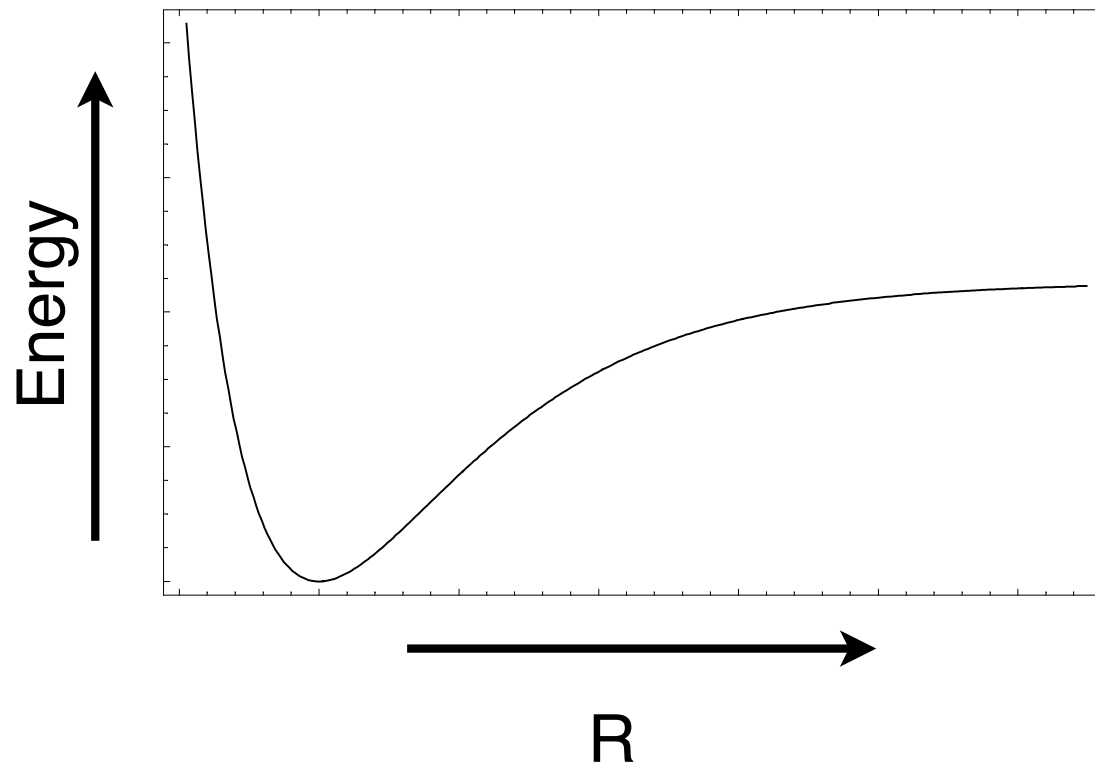
brookite
 $\langle \gamma \rangle = 0.7 \text{ J/m}^2$



anatase
 $\langle \gamma \rangle = 0.5 \text{ J/m}^2$

Forces

- For a given set of atomic positions the atoms will feel a force if not at equilibrium:



- Rule of thumb: $F > 1.0 \text{ eV/\AA}$ is a large force.

- Minimum energy configuration corresponds to zero force
 - Using forces to find the minimum is much more efficient than just using energy alone
- Minimum enthalpy corresponds to zero force and stress
 - Can therefore minimise enthalpy w.r.t. supercell shape due to internal stress and external pressure
 - Can minimize under external pressure to find new structures

- Classically we have the force \mathbf{F} at position \mathbf{R} is determined from the potential energy as

$$\mathbf{F} = -\nabla_{\mathbf{R}} U(\mathbf{R})$$

- Quantum mechanically we therefore expect

$$\mathbf{F} = -\nabla_{\mathbf{R}} \langle E \rangle$$

where

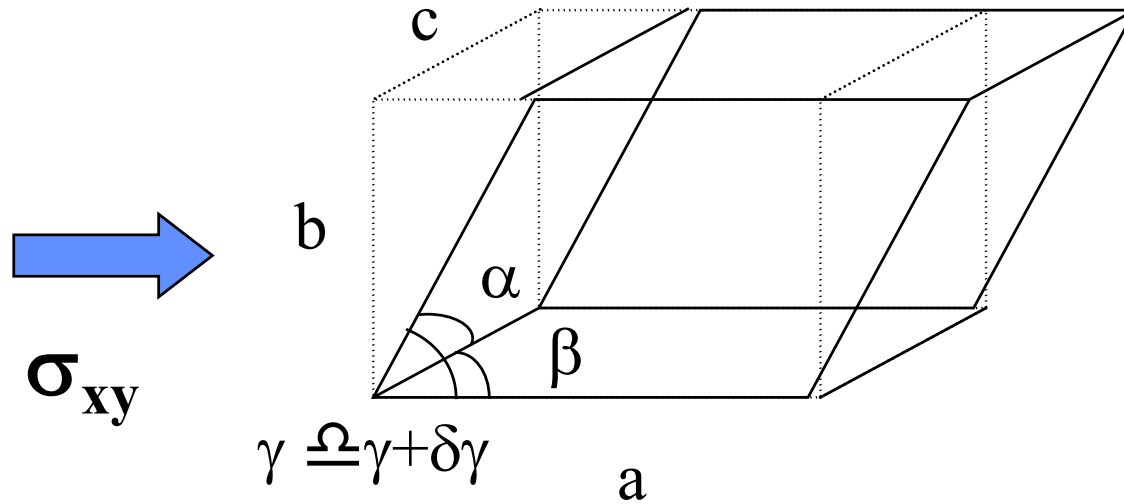
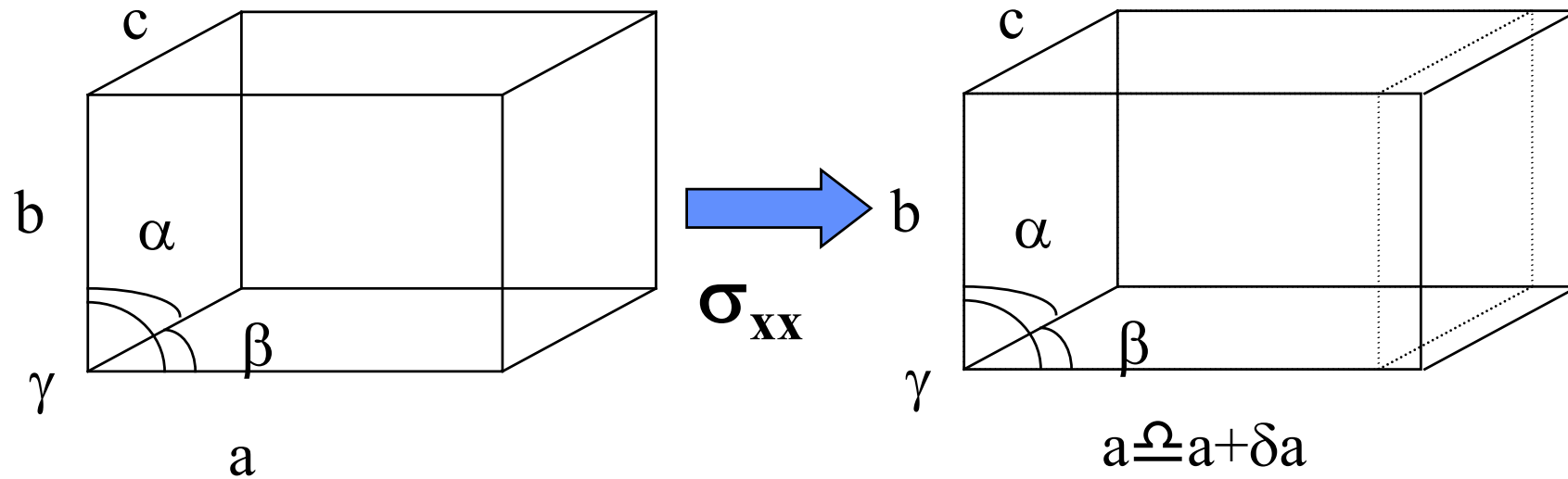
$$\langle E \rangle = \frac{\langle \Psi | H | \Psi \rangle}{\langle \Psi | \Psi \rangle}$$

Stresses

- Whilst force is a simple 3-vector, stress is a rank-2 tensor

$$\boldsymbol{\sigma} = \begin{pmatrix} \sigma_{xx} & \sigma_{xy} & \sigma_{xz} \\ \sigma_{yx} & \sigma_{yy} & \sigma_{yz} \\ \sigma_{zx} & \sigma_{zy} & \sigma_{zz} \end{pmatrix}$$

- Symmetry reduces this to 6 independent components:
 - Diagonal = $\sigma_{xx}, \sigma_{yy}, \sigma_{zz}$ = compression
 - Off-diagonal = $\sigma_{xy}, \sigma_{yz}, \sigma_{zx}$ = shear
- Convention: $\mathbf{P}_{\text{ext}} + \boldsymbol{\sigma} = 0$ at equilibrium
 - i.e. $P > 0$ = compression, $P_{\text{int}} = 1/3 \text{Tr}(\boldsymbol{\sigma})$



NB Much messier if non-orthogonal cell

- If we write the three unit cell vectors **a**, **b**, **c** as the columns of a matrix **h** then the effect of an applied strain ε is to change the shape of the unit cell:

$$\mathbf{h}' = (\mathbf{I} + \varepsilon) \mathbf{h}$$

- We then have the stress tensor σ related to the strain tensor ε by:

$$\sigma_{\alpha\beta} = \frac{1}{V} \frac{\partial E}{\partial \varepsilon_{\alpha\beta}}$$

where V is the volume of the unit cell.

Elastic Constants

- In 1D you know $E = E_0 + \frac{1}{2}k(x - x_0)^2$
where k = elastic constant

- or in terms of dimensionless strain:

$$E = E_0 + \frac{L}{2}k\delta^2$$

- In 3D, we have 4th rank elastic constants tensor but can write as 6x6 using Voigt:

$$E(\epsilon) = E_0 + \frac{V}{2} \sum_{ij} C_{ij} \epsilon_i \epsilon_j + O(3)$$

- which has max 21 unique components

- For certain high-symmetry crystals there are only a few distinct components to c_{ij}
 - Hence can efficiently calculate with the appropriate minimal pattern of strains
 - E.g. in a cubic crystal there are only 3 distinct terms: c_{11} , c_{12} and c_{44}
 - E.g. can choose strain to just study c_{44} :

$$\epsilon = \frac{1}{2} \begin{bmatrix} 0 & \delta & \delta \\ \delta & 0 & \delta \\ \delta & \delta & 0 \end{bmatrix} \quad E = E_0 + \frac{3V}{2} C_{44} \delta^2 + O(3)$$

Summary

- Total energy is VERY useful
 - Changes can determine phase stability, lattice parameters, bond energies, defect energies, surface energies, etc.
- Can also calculate derivatives
 - E.g. forces (change wrt ionic positions)
 - E.g. stress (change wrt cell parameters)
 - E.g. elastic constants (2nd order strain derivative)

- RM Martin, *“Electronic Structure: basic theory and practical methods”*, Cambridge University Press (2004)
- SJ Clark, MD Segall, CJ Pickard, PJ Hasnip, MIJ Probert, K Refson and MC Payne, *“First principles methods using CASTEP”*, Zeitschrift für Kristallographie **220**, 567 (2005)
- MIJ Probert and MC Payne, *“Improving the convergence of defect calculations ...”*, Phys Rev B **67**, 075204 (2003)
- MIJ Probert, *“An ab initio study of xenon retention in α -quartz”*, J.Phys Cond Matt **22**, 025501 (2010)
- PJ Hasnip, K Refson, MIJ Probert, JR Yates, SJ Clark and CJ Pickard, *“Density Functional Theory in the solid state”*, Phil. Trans. R. Soc. A **372**, 20130270 (2014)